Structural and optical studies of tin iron chromium nano oxides

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ABSTRACT

In the present study, we have synthesized Sn $_{0.2}$ Fe $_{0.9}$ Cr $_{0.9}$ O $_{0.2}$ nano oxides. Single phase corundum-like structure and nanophase structure of the as-synthesized sample were confirmed by x-ray diffraction (XRD) and by transmission electron microscope (TEM). The results show that the produced powders have grain size of approximately 32 nm. Using the result of the ultraviolet-visible (UV-Vis) spectrometer, we were able to determine the energy band gaps of 3.1 and 5.2 eV , when nanoparticles are diluted in iron standard solution and in deionised water respectively.

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1 Introduction

Ferric oxide $(\alpha - \text{Fe}_2\text{O}_3)$, chromic oxide $(\alpha - \text{Cr}_2\text{O}_3)$ and their mixed compounds have applications in many fields. They can be used as catalysts for high temperature water gas shift reaction [1] and reduction of NO_x by NH_3 [2], as refractory materials [3] and sensing materials [4].

Their crystal structure is the same as that of corundum (Al₂O₃). α -Fe₂O₃ and α -Cr₂O₃ have different semiconducting and sensing properties. Their magnetic and structural properties are influenced by particle size [5, 6], degree of crystallinity [7], pressure [8] and doping [9, 10].

The effect of metal doping by titanium [11, 12], tin [13, 14], manganese [15], aluminium [16, 17], gallium [18] and indium [19] on the electrical, magnetic and other physical properties of $\alpha - \text{Fe}_2\text{O}_3$ (hematite) has been investigated for some time. The dopant metal ions (tin, titanium) substitute for Fe^{3+} in the corundum-related structure of $\alpha - \text{Fe}_2\text{O}_3$ with the consequent formation of cationic and anionic vacancies. The difference in ionic radius between the dopant metal ion and that for Fe^{3+} will influence the structural characteristics and phase behaviour of the doped system.

The tin-doped $\alpha-\text{Fe}_2\text{O}_3$ system has attracted interest because of its sensing properties for gases such as methane and carbon monoxide [20, 21]. The microstructure of tin-doped $\alpha-\text{Fe}_2\text{O}_3$, prepared by mechanical milling was examined by Jiang et al. [22]. The properties of $\alpha-\text{Cr}_2\text{O}_3$ can also be drastically changed by varying their size and introducing dopants [23, 24, 25]. Literature suggests that introducing dopants into the sensor material is an effective way to improve sensitivity and selectivity [26]. However, literature on gas sensors based on tin-chromium mixed oxides is rarely available [27, 28].



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The substitution of Cr^{3+} by Fe^{3+} to form solid solutions of the type $\alpha-Fe_{2-x}Cr_xO_3$ has been known for some time [29]. Although there have been several studies of the $\alpha-Fe_{2-x}Cr_xO_3$ [21, 29, 30] system, there appears to have been little activity in the doping of these materials by tin. Berry et al. [31] initiated a study of tin-doped oxides of the type $\alpha-Cr_2O_3$ and $\alpha-(FeCr)_2O_3$ and reported on their structural properties.

Therefore, study about effects of these dopants on the structure and optical properties of $(FeCr)_2O_3$ are very important for future applications. In this paper we report the synthesis of $Sn_{0.2}Fe_{0.9}Cr_{0.9}O_3$ using a simple low cost chemical route and study the structure, characterisation and optical absorption properties.

2 Materials and Methods

Single-phase corundum compounds were prepared by mixing an aqueous solution of tin (II) chloride dehydrate with an aqueous solution of chromium (III) chloride hexahydrate and iron (III) chloride hexahydrate. Excess aqueous ammonia was added to the mixtures and the solutions boiled in a stirred pressure reactor for 3 hours. The resulting precipitate was filtered, washed with deionized water until no chloride ions were detected by the silver nitrate standard solution. The precipitate was washed with 95% ethanol and dried under an infrared lamp before calcinations in air at 600° C for 12 hours. The products were characterized by X-ray diffraction (XRD) using $C_{o}K_{\alpha}$ radiation ($\lambda=1.7903\,\text{Å}$) by a Phillips diffractometer (type: PW1710). A Jeol model JEM-1010 transmission electron microscopy (TEM) was used to obtain the morphology of the samples.

Nano-material (powder) were prepared for spectrophotometric analysis using two different solvents, namely the water de-ionized (H_2O) from UniLab SAARS6725000LL and Iron standard solution, traceable to SRM from NIST Fe(NO_3)3 in HNO3, (0.5 mol/l). Each substance was accurately weighted using an electronic balance, each sample of 50.0 mg of powder was diluted in 10 ml of solvent. The UV-6300 PC Spectrophotometer (UQE1706002) with a wavelength range of 190-1100 nm at room temperature, was used to record the optical spectrum of the prepared sample. The UV-Vis Analyst-0001 software was used to display the spectrum.

3 Results and discussion

The XRD measurements were scanned in the range $20^{\circ} \le 2\theta \le 80^{\circ}$ in steps of 0.01/s. The XRD pattern of the as prepared sample of Sn_{0.2}Fe_{0.9}Cr_{0.9}O₃ is shown in Figure 1.

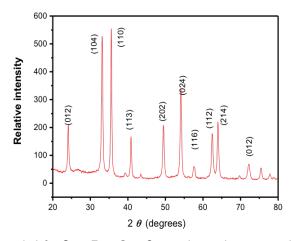


Figure 1: XRD patterns recorded for $Sn_{0.2}Fe_{0.9}Cr_{0.9}O_3$ synthesized in a stirred pressure reactor at $100^{\circ}C$.

The results are similar to those recorded from tin-doped $\alpha-\text{Fe}_2\text{O}_3$, $\alpha-\text{Cr}_2\text{O}_3$ and tin-doped $\alpha-(\text{Fe}\text{Cr})_2\text{O}_3$ [20, 23]. The refinement of XRD patterns was then performed by a Rietveld analysis according to an identical model to that was previously used to refine the diffraction data of tin-doped $\alpha-\text{Fe}_2\text{O}_3$ and $\alpha-\text{Cr}_2\text{O}_3$ [20]. The structure of $\alpha-\text{Fe}_2\text{O}_3$ has Fe^{3+} ions distributed in an ordered fashion in 2/3 of the octahedral sites within a framework of hexagonally close-packed O^{2-} ions. Chains of face sharing octahedral are directed along the c axis, and the Fe^{3+} ions within each chain form pairs as shown in Figure 2.

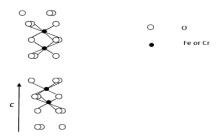


Figure 2: Linking of FeO₆ or CrO₆ octahedra along c in α -Fe₂O₃ or α -Cr₂O₃.

It was shown [20] that for such structures, the data cannot be fitted to a simple model in which Sn^{4+} substitute for Fe^{3+} with charge balance being achieved by an appropriate number of cations vacancies. The model that we use involves tin ions substituting on two of the octahedral Fe^{3+} sites in corundum related $\alpha-\mathrm{Fe}_2\mathrm{O}_3$. Elimination of Fe^{3+} cations from the octahedral adjacent to the Sn^{4+} creates two additional octahedral sites that do not involve face sharing and which are therefore amenable to occupation by additional Sn^{4+} ions. The structure is formed by defect clusters comprising a chain of three tin ions which all avoid face-sharing repulsions as shown in Figure 3. The cluster is electrically neutral since 4 Fe^{3+} ions are replaced by 3 Sn^{4+} ions. In the final stage of the refinements, the cation site occupancies were constrained in accordance with the model.

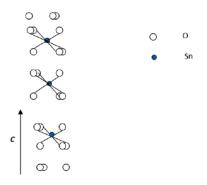


Figure 3: Structural model involving the substitution of 4 Fe³⁺ or 4 Cr³⁺ ions by 3 Sn⁴⁺ ions.

The measured data and refinement results for $Sn_{0.2}Fe_{0.9}Cr_{0.9}O_3$ are shown in Figure 4, where the black circles indicate the experimental data, red solid line indicates the Rietveld refinement fit by Fullprof analysis software. The green curve at the bottom reveals the difference between the fit and the experimental data.

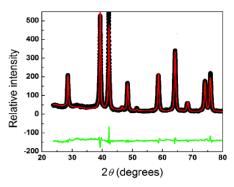


Figure 4: Experimental powder X-ray diffractogram Sn_{0.2}Fe_{0.9}Cr_{0.9}O₃ of (black symbols), Rietveld refinement fit (solid red line) and difference curve (green line).

The complete set of final refined parameters are shown in Table 1.

Table 1: Refined Structural Parameters.

Atom	x/a	y/b	x/c	$U_{ISO} \times 100/\text{Å}^2$	Occupancy
CR1	0.00000	0.00000	0.3463(2)	0.500(0)	9.016(2)
Fe	0.00000	0.00000	0.3463(2)	0.500(0)	1.696(2)
Sn 1	0.00000	0.00000	0.3463(2)	0.500(0)	0.447(3)
Sn 2	0.00000	0.00000	0.00000	0.00000	0.500(0)
01	0.3217(1)	0.00000	0.2500(0)	0.500(0)	18.000(0)
R_{exp} (%)	14.4	0.00000	R_{exp}	9.5	,

The structural parameters show little deviation from those of ideal $\alpha-\text{Fe}_2\text{O}_3$ and $\alpha-\text{Cr}_2\text{O}_3$ [31] which suggests that the defect clusters can be incorporated into the corundum structure with very little lattice strain. Direct measurement of the particle size and powder morphology for the as-prepared sample were also performed by transmission electron microscopy (TEM). The TEM image in Figure 5 shows that the sample consists of small nano-sized particles.

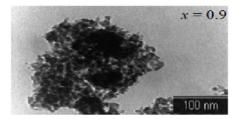


Figure 5: TEM micrograph for the $Sn_{0.2}Fe_{0.9}Cr_{0.9}O_3$ sample.

The calculated D_{XRD} and measured D_{TEM} values appear to be in good agreement. The lattice parameters, the average particle diameters as calculated from XRD data (D_{XRD}) and observed by TEM measurements (D_{TEM}) are shown in Table 2.

Table 2: Particle size (D_{XRD}) , (D_{TEM}) and lattice parameters for as-prepared $Sn_{0.2}Fe_{0.9}Cr_{0.9}O_3$ sample.

	a = b	c	V	D _{XRD}	D _{TEM}
X	(Å)	(Å)	$(Å^3)$	(nm)	(nm)
	$\pm 10^{-3}$	$\pm 10^{-3}$	$\pm 10^{-3}$	± 0.4	± 2
0.9	4.981	13.631	291.783	31.5	32

The UV-Vis transmittance and absorbence spectra of the as prepared samples are shown in Figure 6 and Figure 7.

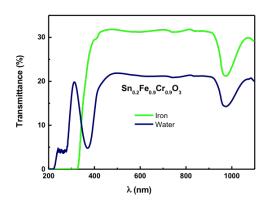


Figure 6: UV-vis transmittance spectrum of $Sn_{0.2}Fe_{0.9}Cr_{0.9}O_3$ nanoparticles.

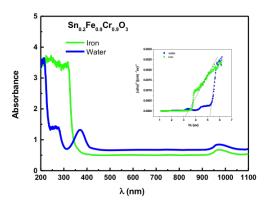


Figure 7: UV-vis absorbance spectrum of $Sn_{0.2}Fe_{0.9}Cr_{0.9}O_3$ nanoparticles. Inset shows the determination of optical bandgap energies for the respective nanoparticles.

The transmittance spectrum of the prepared $Sn_{0.2}Fe_{0.9}Cr_{0.9}O_3$ shows that, in both the UV (400 nm) and visible (400-800 nm) regions the sample is having high transparency and its transmittance is lower for the sample diluted in water. The absorbance spectrum of the sample shows that the absorption edge shifted towards longer wavelengths (red shifted) from 250 to 350 nm from the sample diluted in water to the one with the Fe

concentration. The optical band gap energy (E_g) of the as-synthesized nanoparticles is obtained from the UV-Vis spectra by using a well-known Tauc's relation [32]:

$$(\alpha h v) = A(h v - E_q)^n, \tag{1}$$

where α is the absorption coefficient, A is a constant, E_g is the band gap energy of the material and exponent $n=\frac{1}{2}$ for direct transition. The value of the absorption coefficient can be determined by the following equation [33]:

$$\alpha = 2.303 \left(\frac{\text{absorbance}}{t} \right), \tag{2}$$

where 't' is the thickness of the sample. The curves are plotted between $(\alpha h v)^2$ versus (hv) and extrapolating of the linear portions of the curves to the hv axis gives $E_g=3.1$ eV for the sample diluted in iron solvent which is in agreement with the reported values by Huagiang, et al., [34] and 5.2 eV for the sample diluted in deionised water as shown in the inset of Figure 7.

4 Conclusion

 $Sn_{0.2}Fe_{0.9}Cr_{0.9}O_3$ fine powders have been successfully synthesized by hydrothermal process at low temperature ($100^{\circ}C$) and annealed at $600^{\circ}C$. The XRD patterns indicate that the sample has single phase. The crystallite size calculated from XRD data shows good agreement with the particle size obtained by TEM. The absorbance spectrum of the prepared sample shows that the absorption edge shifted towards longer wavelengths (red shifted) from 250 to 350 nm when the sample is diluted in iron standard solution with an energy gap of 3.1 eV.

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